
Pulse White Paper

on the Sn Whisker Growth Propensity of
Pulse Electronics Component Terminals:

- Tin-Plated Leadframes
 - Solder Dipped Tin Plated Pins
 - Solder Dipped Copper Wire Leads
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I. INTRODUCTION

World wide environmental regulations have required electronics manufacturers to adopt lead-free and environmentally friendly processes and materials in the construction of their components. Until the recent enactment of the EU and China reduction of hazardous materials (RoHS) legislations, the use of lead used in solders to make internal connections and lead-frame platings was widespread. Pure Sn (tin) and Sn rich alloys and solders have become viable options due to their compatibility and relative low cost.

The transition to these Pb-free materials has presented technical challenges and quality and reliability concerns. The major concern is that of the growth of Sn filaments or whiskers that have the potential to cause malfunctions by shorting electrical circuits. This was first observed in 1947.

This paper will address the concerns Pulse customers have regarding the Sn whisker propensity of Sn plated and Sn dipped surface finishes.

II. BACKGROUND

There is much interest and ongoing research in Sn whisker growth, but due to the complexity and multiple contributing factors involved; there is no one single model that can describe the phenomenon entirely. Despite the differing opinions it is widely accepted that compressive forces drive Sn whisker formation. The forces responsible for Sn whisker growth can be grouped into two broad categories of factors:

- 1) The sources of the compressive stress and;
- 2) The mechanism by which the stress is relieved.

The factors that contribute to the formation of compressive stresses are mechanical, thermal and chemical: 1) Formation of an intermetallic compound (ImC) Sn_6Cu_5 . 2) Coefficient of thermal expansion (CTE) mismatch between the base metal substrate and the plating layer. 3) Induced mechanical stress.

The factors that affect the manner in which the stress is released are: 1) Thickness of the Sn layer. 2) Sn grain morphology. 3) Thickness of the surface SnO_x . 4) Surface imperfections. Both the stress sources and relief mechanism will be further discussed.

SOURCES OF COMPRESSIVE STRESSES

A. *Intermetallic Compound Growth*

Leadframe base metal composition plays an important role in the components Sn whisker propensity. A commonly used alloy in leadframe construction is phosphor bronze. Phosphor bronze alloys composition is 60% Copper, 28% Zinc, 0.3% Phosphorus and other trace metals. At the base metal and plating layer interface, Copper is the primary diffusing species into Sn. Sn diffusion into Cu is negligible. The diffusion of Cu shown as upward arrows in Fig. 1 from the Cu-alloy substrate into the Sn layer forms two inter-metallic compounds, Sn_6Cu_5 and Cu_3Sn at the Cu-Sn boundary as shown in Fig. 1. Lower initial thermal conditions kinetically favor the formation of over Sn_6Cu_5 over Cu_3Sn . Sn_6Cu_5 has a lower density than Cu_3Sn and the surrounding Sn. At ambient temperature Sn_6Cu_5 is the preferred crystal and has a scallop

morphology. This type of structure occupies a greater Molar volume than the Cu_3Sn , which form a layer structure and the Sn it displaces, thus exerts compressive forces shown as horizontal

arrows in Fig. 1 upon the Sn overplating. The system relieves these stresses by the protrusion of a Sn whisker at the Sn grain boundary on the surface.

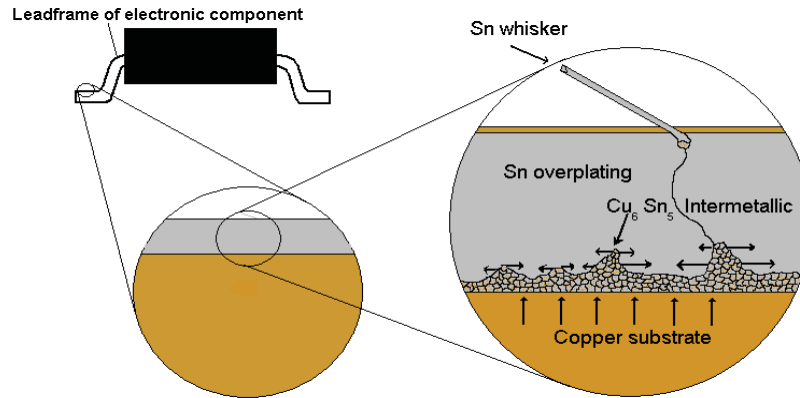


Fig 1. Schematic diagram of the layered Sn-Cu alloy structure of a barrier-less lead-frame.

ImC compressive stress plays a prominent role in copper rich alloys, such as phosphor bronzes. Phosphor bronze based leadframes are thus prone to Sn whisker if left unmitigated.

B. Stress Induced by Mismatching Coefficients of Thermal Expansion

Another common but less used material in leadframe construction is Alloy 42. This alloy contains 42% Nickel and the remainder Iron with traces amounts of 0.2% Silicon and 0.4% Manganese. During thermal cycling, the Sn layer experiences tensile and compressive forces. Due to differences CTE's the metal substrate will prevent the Sn layer from expanding. This will generate compressive stress on Sn layer.

Alloy 42 leadframes are not susceptible to ImC growth, but are more sensitive to dimensional distortions due differences in CTE's and in turn produce compressive stress, as shown in Fig. 2 below. This difference in CTE's is shown by the red arrows.

Phosphor bronze alloys will also experience compressive stress in addition to that imparted by growth of an ImC. Sn and Alloy 42 have corresponding CTE values of $22.0 \times 10^{-6}/^{\circ}\text{C}$ and $4.0-4.3 \times 10^{-6}/^{\circ}\text{C}$. It is difficult to find agreeing values for CTE's for ImC's in the literature due to the difficulty in obtaining homogeneous samples in sufficient quantities for testing. Therefore values have been omitted. For these reasons Sn-Alloy 42 systems are benign to Sn whiskering if temperature cycling and storage conditions are kept to a minimum and close to ambient temperature.

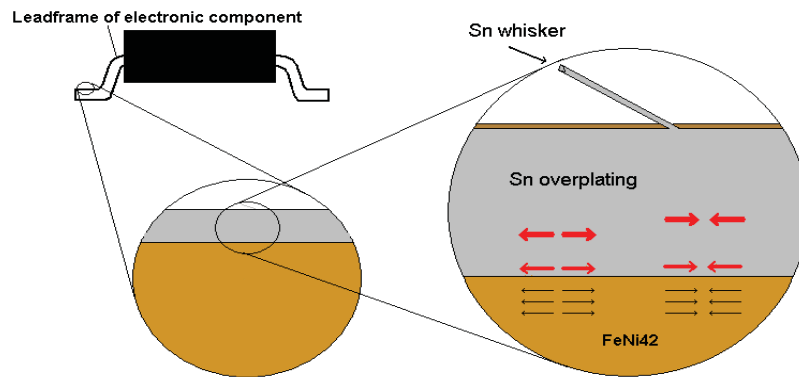


Fig 2. Schematic diagram of the layered Sn-Alloy 42 structure of a barrier-less lead-frame.

C. Applied Mechanical Stress

The bending, twisting and clamping forces applied to the leadframe during manufacture, assembly of product and normal operating conditions will produce compressive stresses at the point of contact and also in bends and joints. This is a concern, but to a lesser degree than the previous mentioned ones.

D. Additives and Impurities in Sn Layer

Certain metals intentionally added or present as impurities can decrease the Sn whisker propensity by lowering the CTE of the alloy or increase it if they readily oxidize and increase compressive stresses.

The presence of copper in the SAC alloy has the effect of lowering the CTE of tin from -22 to $-14 \times 10^{-6}/^{\circ}\text{C}$, while forming a barrier between the tin plates.

Some rare earth elements such as Cerium and Lutetium present in trace amounts have been reported to cause extensive whiskering since they readily oxidize and increase compressive stresses.

It has been reported that the carbon content of the plating solution is a factor. It is believed that the plating solution used to electro-deposit the Sn grains comprising the Sn layer will form finer grains with increasing carbon concentration. There is some controversy as it is also possible to make fine Sn grains with low carbon content solutions.

STRESS RELAXATION MECHANISMS

Stress relaxation factors play a key role in the whisker surface density, length and morphology of the Sn whisker. Whiskers have been reported to grow as much as a few hundred micrometers

with an approximate diameter of three micrometers. The stress relaxation factors are further described.

A. Sn Layer Thickness

Sn whisker surface density is inversely proportional to the thickness of the Sn layer. More whiskers are found on thin finishes than thicker ones. The reason is compressive stresses propagate and remain more focused to the surface from the ImC layer thru shorter distances. Thicker finishes typically have increased wettability problems so there is an optimum layer thickness between 5-10 μm .

B. Sn Grain Morphology

As the name implies, Sn whiskers are long filaments protruding from the plating surface. Sn whiskers can also be irregular or nodular in appearance as shown in Fig 3. This type of whisker is called the hillock-type.

The length of the Sn whisker is believed to depend on the morphology and orientation of the Sn grain from which it emerges. It has been observed that rougher larger Sn grains (matte) will produce hillock-type whiskers and smaller (bright) grain finishes will produce filaments. A possible explanation is that compressive stress is partially countered by tensile stress generated by bulk diffusion of Cu thru the larger voids in between larger grain sizes. Filaments will erupt from ternary grain boundaries while hillocks will erupt from entire grains.

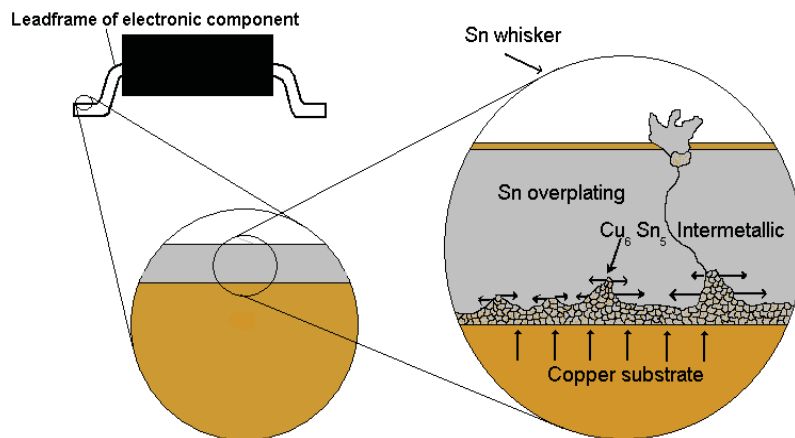


Fig 3. Schematic diagram of the layered Sn-Cu alloy structure of a barrier-less lead-frame.

C. Sn Oxide Layer and Surface Imperfections

Sn plated surfaces readily form an oxide layer, SnO_x . The presence of this layer is critical for whisker formation. This layer acts as a barrier that prevents a vacancy gradient flux from being established that originates from the surface down to the ImC layer. Sn creep induced by compression to the surface thru these vacancies is blocked and compressive stress builds and will eventually cause a whisker to erupt wherever there is a weak spot. Metals such Cu and Fe that do not form oxide layer have not been reported to whisker.

III. TIN WHISKER MITIGATION MEASURES

Scientists, engineers and researchers from eight leading manufacturers of electronics goods have formed a consortium called the international Electronics Manufacturing Initiative (iNEMI) Tin Whisker Group. This group's task is to make recommendations on Pb-free finishes. Pulse has adopted the Tin whisker mitigation recommendations made by the iNemi. These accepted mitigation measures are detailed in the following section.

A. Minimal Sn Overplating Layer Thickness

A minimal 2 μ m thick Sn layer is recommended for Ni barrier containing finishes. An 8 μ m thick layer is recommended for barrier-less finishes.

B. Ni Underplating Barrier Layer

The addition of an underplating layer decreases the compressive stress upon the overplated Sn layer by blocking the diffusion of Cu and the formation of ImC's, thus mitigating Sn whisker formation. There is a second mechanism by which Ni underplatings mitigate Sn whisker formation. The Sn in the overplating layer diffuses into Ni to form the intermetallic Ni₃Sn₄. This leaves vacancies in the Sn layer that that establish tensile stress.

C. Sn Grain Size

As has been previously mentioned, large Sn grains (matte) in the size range of 1-5 μ m will form hillock-type protrusions on the Sn surface layer as opposed to filaments. Matte Sn finishes are the only type used.

D. Sn0.7Cu Solder

The addition of a coating of solder has proven to be highly effective in mitigated Tin whisker growth on Tin plated finishes. Pulse standard lead free solder is Cu saturated Sn solder. Sn0.7Cu is less aggressive on copper wires compared to 100% Sn solder, increasing the reliability of the small wire size to pin solder terminations

IV. PULSE COMPONENT PLATFORMS

Pulse manufactures a large variety of passive components. Many of them have been re-engineered or converted over to be Pb-free RoHS compliant. The concern faced by Pulse customers is with the components containing these Pb-free materials. The impacted products are of three class types: Tin plated leadframes, solder dipped tin plated pins and solder dipped copper wire leads. The Sn whisker mitigation measure is unique to each family of components and will be further detailed in the following sections.

A. *Tin Plated Leadframes*

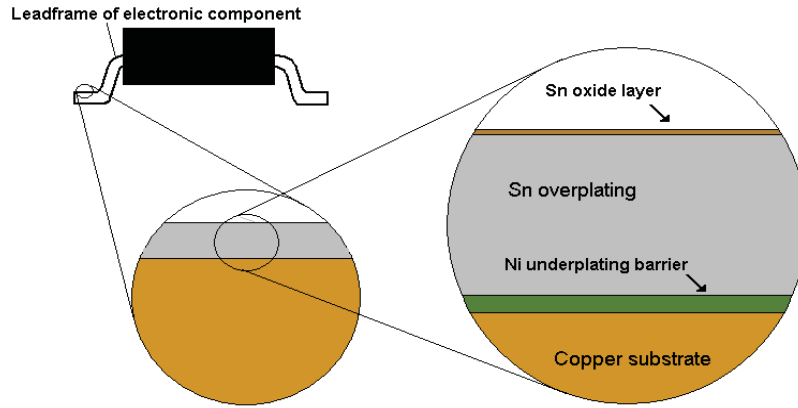


Fig 4. Schematic diagram of the layered Cu-Alloy structure of a Ni barrier containing lead-frame.

B. *Solder Dipped Tin Plated Pins*

Tin plated pins are dipped into hot Sn0.7Cu solder. This, in addition to the above, further mitigates against the growth of Tin whiskers

C. *Solder Dipped Copper Wire Leads*

Non plated copper wire Leads are dipped into hot Sn0.7Cu wire. The nominal thickness of this layer is 50µm. The thickness of this layer far exceeds the 8µm minimum recommended by the iNEMI Tin Whisker Group

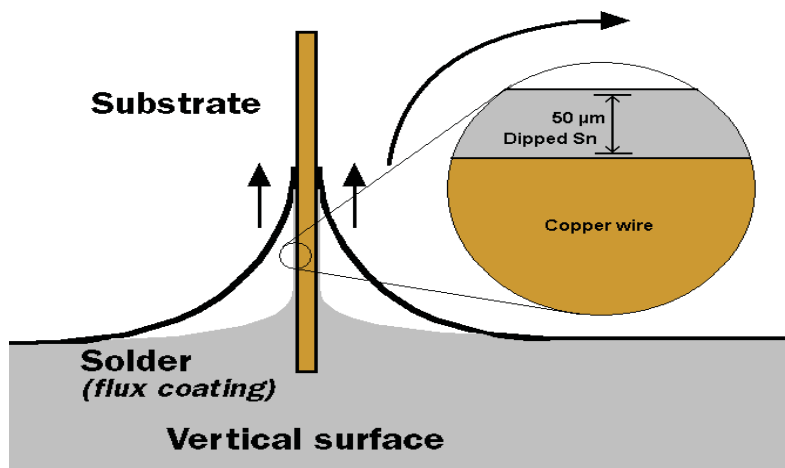


Fig 5. Schematic diagram of the Sn dipped wire surface structure.

V. SUMMARY

Pulse maintains strict quality control measures to assure the purity of the materials used in the manufacture of our components. Pulse adheres to the iNemi consortia recommendations on Tin Whisker Mitigation recommendations, such as minimum plating thickness, Nickel Barrier layer, Matte Tin grains and post solder dipping, where feasible

Pulse has performed Tin Whisker testing based on the industry standard JESD22A121 to demonstrate that the Sn whisker propensity of its components is effectively mitigated. Please contact your sales representative for more information.

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